

Original article

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**EFFECT OF ARGON ION BOMBARDMENT  
ON THE COMPOSITION, ELECTRONIC STRUCTURE  
AND PHYSICAL PROPERTIES OF CADMIUM FLUORIDE**

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**Abstract.** In the paper, the effect of bombardment with Ar<sup>+</sup> ions on the composition, electronic and crystal structure of the surface layers of bulk single crystal samples and CdF<sub>2</sub>(111) films has been studied using the methods of Auger electron and ultraviolet photoelectron spectroscopy, high-energy electron diffraction and recording the angular dependences of the reflectance factor of inelastically reflected electrons. The effect of this bombardment on the density of states of valence electrons and energy band parameters of CdF<sub>2</sub> was investigated for the first time. The degree of disorder of CdF<sub>2</sub> into components and evaporation of fluorine from the surface layers was established to depend on the energy and dose of Ar<sup>+</sup> ions. The complete evaporation of F in the form of a diatomic gas was shown for the first time to be observed in the energy range of 1 – 2 keV at a saturation dose.

**Keywords:** epitaxial layer, heterostructures, ion bombardment, Auger spectrum, photoelectron spectrum, disordered layer, electron density of state

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Научная статья

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**ВЛИЯНИЕ БОМБАРДИРОВКИ ИОНАМИ АРГОНА НА СОСТАВ,  
ЭЛЕКТРОННУЮ СТРУКТУРУ И ФИЗИЧЕСКИЕ СВОЙСТВА  
ФТОРИДА КАДМИЯ**

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**Аннотация.** В работе изучено влияние бомбардировки ионами аргона  $Ar^+$  на состав, электронную и кристаллическую структуру поверхностных слоев объемных монокристаллических образцов и пленок фторида кадмия  $CdF_2(111)$ . Для этого использованы методы оже-электронной и ультрафиолетовой фотоэлектронной спектроскопии, дифракции быстрых электронов и регистрация угловой зависимости коэффициента отражения неупругоотраженных электронов. Впервые изучено влияние указанной бомбардировки на плотность состояния валентных электронов и энергетические зонные параметры  $CdF_2(111)$ . Установлено, что степень разупорядочения  $CdF_2$  на составляющие и испарение фтора с поверхностных слоев зависит от энергии и дозы ионов  $Ar^+$ . Впервые показано, что полное испарение фтора в виде двухатомного газа наблюдается в области энергий 1 – 2 кэВ при дозе насыщения.

**Ключевые слова:** эпитаксиальный слой, гетероструктуры, ионная бомбардировка, оже-спектр, фотоэлектронный спектр, неупорядоченный слой, плотность электронных состояний

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## Introduction

The great interest in epitaxial fluoride layers is associated both with the unique properties of the latter and with the wide potential possibilities of their application in opto- and microelectronics [1 – 13]. In particular, metal fluorides are widely used in the creation of special semiconductor – dielectric – semiconductor (SDS) structures in the three-dimensional integrated circuits. Of particular interest are  $CdF_2/Si(111)$  heterostructures with a  $CaF_2$  buffer layer [9, 14, 15]. The minimum thickness of the  $CaF_2$  buffer layer was 0.9 nm [5]. In this case,  $CaF_2$  plays the role of a barrier layer for the chemical reaction between  $CdF_2$  and Si substrates [9]. At the same time, trivalent germanium turned out to be the most promising for doping  $CdF_2$  [16].

Single-crystalline cadmium fluoride is a solid dielectric that can be converted into a semiconductor by doping with donor impurities and subsequent heating in a reducing atmosphere [16 – 18].

In Refs. [19, 20], the energy position of the levels of rare earth (RE) elements in the band diagram of  $BaF_2$  and  $CdF_2$  crystals was determined. The role of  $RE^{3+}$  and  $RE^{2+}$  ions in the carrier capture, luminescence, and the formation of radiation defects was assessed. It was shown that the significant difference in the luminescent properties of  $BaF_2:RE$  and  $CdF_2:RE$  was due to the position of excited energy levels in the band diagram of the crystals. In Ref. [21], Shubnikov – de Haas oscillations and a quantum staircase of the Hall resistance were discovered in a  $p$ - $CdF_2$  quantum well limited by  $Cd_{1-x}B_xF_{2-x}$   $\delta$  barriers on the  $n$ - $CdF_2$  surface. Thanks to the low effective



mass of two-dimensional holes, the observation of the quantum Hall effect became possible at room temperature.

Studying the influence of various external influences, especially ion bombardment, on the composition, structure and physical properties of fluorides is of both fundamental and applied interest. In recent years, we have thoroughly studied the effect of ion bombardment on the composition, electronic and crystal structure, emission and optical properties of dielectric films and samples [12 – 27]. However, to date, the effect of low-energy ion bombardment on the composition and properties of  $\text{CdF}_2$  films has been practically unstudied.

In this work, changes in the composition, electronic and crystal structure of  $\text{CdF}_2$  (111) upon bombardment with  $\text{Ar}^+$  ions were studied for the first time.

### Experimental methods

The subject of research was a single-crystal sample of  $\text{CdF}_2$  (111) with a thickness of about 0.5 mm and molecular beam epitaxial (MBE) films of  $\text{CdF}_2/\text{Si}(111)$  with a thickness of about 500 Å. Before ion bombardment, the samples under study were degassed at  $T \approx 1000$  K for 3 hours in a vacuum (pressure  $P \approx 10^{-7}$  Pa). The elemental and chemical compositions of the samples were determined by Auger electron spectroscopy (AES). The degree of amorphization of the  $\text{CdF}_2$  film upon bombardment with  $\text{Ar}^+$  ions and its crystallization during annealing, the type and parameters of the lattice were studied by high-energy electron diffraction (HEED) method and by measuring the angular dependences of the reflectance factor  $\eta$  of inelastically reflected electrons. To study the density of state of valence electrons and determine the parameters of energy bands, the method of ultraviolet photoelectron spectroscopy (UPS) was used. All measurements were carried out after the target was cooled to room temperature, in a vacuum with a pressure of at least  $10^{-7}$  Pa. The choice of the (111) plane has been due to the fact that the  $\text{CdF}_2(111)$  surface has the lowest free energy ( $E_{\text{CdF}_2} \approx 5 \cdot 10^{-7} \text{ J} \cdot \text{cm}^{-2}$ ,  $E_{\text{Si}} \approx 1.35 \cdot 10^{-4} \text{ J} \cdot \text{cm}^{-2}$  and is atomically smooth.

### Experimental results and their discussion

The Auger spectrum of a well-cleaned  $\text{CdF}_2(111)$  surface is shown in Fig. 1. It can be seen that the  $\text{CdF}_2$  surface contains mainly an impurity of oxygen atoms with a concentration of no more than 1 at.%. The  $\text{CdF}_2$  film surface has high crystalline perfection and an atomically smooth surface with a  $(1 \times 1)$  structure reflection high-energy electron diffraction (RHEED) image (see inset in Fig. 1).

An analysis of the dependence of the intensity  $I_F$  of the Auger peak from fluorine at an energy of 646 eV on the irradiation dose to the  $\text{CdF}_2(111)$  surface bombarded by  $\text{Ar}^+$  ions with different energies  $E_0$  (see Fig. 2) allows us to conclude the following. The intensive desorption of fluorine from the surface of  $\text{CdF}_2$  occurs, starting from the irradiation dose  $D = (1 - 5) \cdot 10^{13} \text{ cm}^{-2}$  and up to  $D = 10^{16} \text{ cm}^{-2}$ ; the rate of decrease in the intensity  $I_F$  depending on the energy  $E_0$ . In particular, for  $E_0 = 0.5$  keV at  $D = 5 \cdot 10^{16} \text{ cm}^{-2}$ , the intensity of the Auger peak corresponds to a minimum, but  $I_F$  is not equal to zero even at  $D = 10^{17} \text{ cm}^{-2}$ , and thus  $D = (4 - 5) \cdot 10^{16} \text{ cm}^{-2}$  is the saturation dose  $D_s$  for  $E_0 = 0.5$  keV. For  $E_0 = 1.0$  keV, a decrease in  $I_F$  to zero is observed at  $D = (6 - 8) \cdot 10^{16} \text{ cm}^{-2}$ . A decrease in  $I_F$  to zero occurred up to  $E_0 = 2$  keV. At  $E_0$  more than 2 – 3 keV, the  $I_F$  value, even at dose  $D$  more than  $10^{17} \text{ cm}^{-2}$ , was above zero. Apparently, at high energies of  $\text{Ar}^+$  ions, the decomposition of  $\text{CdF}_2$  predominantly occurs in the surface layer and complete evaporation of fluorine atoms from these layers does not occur. Or, the evaporation of Cd and  $\text{CdF}_2$  as a whole may begin simultaneously with the evaporation of fluorine atoms.

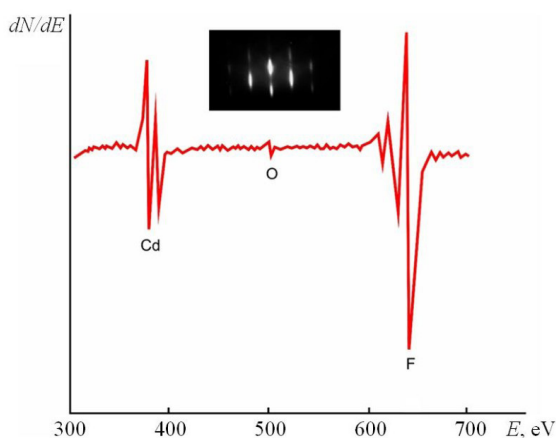


Fig. 1. The Auger spectrum and RHEED images (inset) of the pure  $\text{CdF}_2(111)$  surface

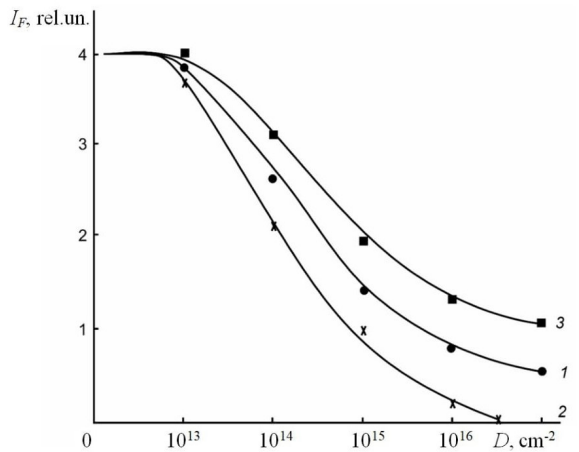


Fig. 2. Plots of the Auger F peak intensity (at  $E = 646$  eV) versus the Ar<sup>+</sup> irradiation dose  $D$  for CdF<sub>2</sub> bombarded by Ar<sup>+</sup> ions with different energy values;  $E_0$ , keV: 0.5 (1), 1.0 (2), 2.5 (3)

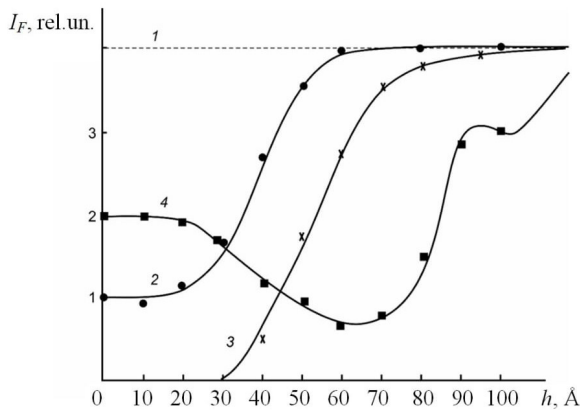


Fig. 3. The depth distribution profiles of F atoms from CdF<sub>2</sub> bombarded by Ar<sup>+</sup> ions at  $D = D_s$  with different energy values;  $E_0$ , keV: 0.0 (1) 0.5 (2), 1.0 (3), 5.0 (4)

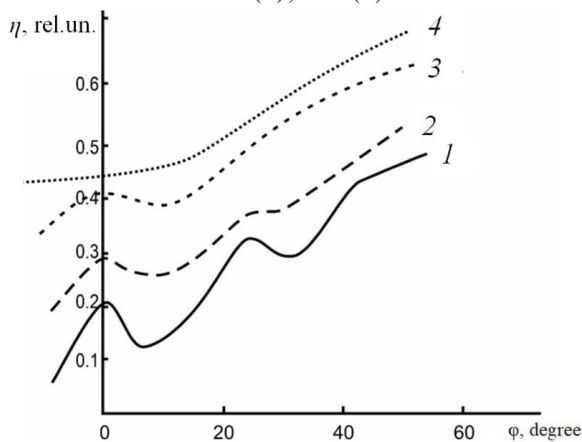


Fig. 4. The dependence of  $\eta$  on the angle  $\phi$  (the angle of incidence of the primary beam) for amorphous CdF<sub>2</sub> films on Si(111) substrates; the film thickness values, Å: 10 (1), 20 (2), 40 (3), 50 (4).  $E_p = 0.8$  keV

To answer this question, we studied the dependence of  $I_F$  on the depth  $h$  of a CdF<sub>2</sub> layer bombarded with Ar<sup>+</sup> ions with different energies at  $D = D_s$  (Fig. 3). The plots in Fig.3 show that the intensity of the  $I_F$  peak decreases sharply (by  $\sim 4$  times) at  $E_0 = 0.5$  keV, and it practically does not change until the layer depth  $h = 20 - 80$  Å, that corresponds to the projected range of Ar<sup>+</sup> ions.

Apparently, all F atoms in the form of diatomic gas F<sub>2</sub> evaporate from these layers. In the region  $h \approx 25 - 50$  Å, the  $I_F$  concentration increases and, starting from  $h \approx 50$  Å, the stoichiometric composition of CdF<sub>2</sub> is completely established. However, further studies showed that the CdF<sub>2</sub> layers were highly disordered to a depth of  $h \approx 130 - 150$  Å. In the case of  $E_0 = 1.0$  keV, the surface layers of the cadmium fluoride are completely decomposed into components to a depth of  $h = 30 - 40$  Å, and almost all F atoms evaporate from these layers, and hence an amorphous cadmium film with a thickness of  $d = 30 - 40$  Å is formed on the surface (see curve 3 in Fig. 3). When CdF<sub>2</sub> is bombarded by Ar<sup>+</sup> ions with  $E_0 = 5$  keV, the greatest decomposition occurs at the depth of the projected range of Ar<sup>+</sup> ions ( $h \approx 60 - 70$  Å). Apparently, most of the fluorine atoms go into vacuum, and the other part diffuses deep into the target. Therefore, the concentration of F increases significantly at a depth of  $h = 80 - 100$  Å.

It is known that the thickness of disordered layers of a single crystal under ion bombardment is very difficult to determine experimentally. In this work, the depth of such layers was assessed by investigation of the angular dependences of the reflectance factor  $\eta$  of inelastically scattered electrons at the different primary electron energies  $E_p$ .

The dependences of  $\eta$  on the angle of incidence of the primary beam on the surface, for the Si(111) system with amorphous CdF<sub>2</sub> films of various thickness values are presented in Fig. 4. They were recorded at  $E_p = 0.8$  keV. It can be seen that the main Si(111)  $p_p$  peak is completely smoothed out at a film thickness of about 50 Å. A similar method was used to determine the thicknesses of the CdF<sub>2</sub> films at which the main Si peak was smoothed out in the range  $E_p = 1 - 10$  keV. The results are given in Table.

Using the data from Table, the thickness of disordered layers  $d_p$  in the CdF<sub>2</sub>(111) was estimated using the condition when bombarded with Ar<sup>+</sup> ions with different energies  $E_0$  at doses  $D = D_s$  (Fig. 5). This plot shows that

**The dependence of the thickness of  
the amorphous CdF<sub>2</sub>/Si(111) films on  
the primary electron energies E<sub>p</sub>**

E <sub>p</sub> , keV	0.8	1.0	3.0	5.0	10
d, Å	50	80	200	350	500

Footnote: *d* values were found under the condition when the main peak on the curve  $\eta(\varphi)$  was smoothed out at a given value of energy  $E_p$ .

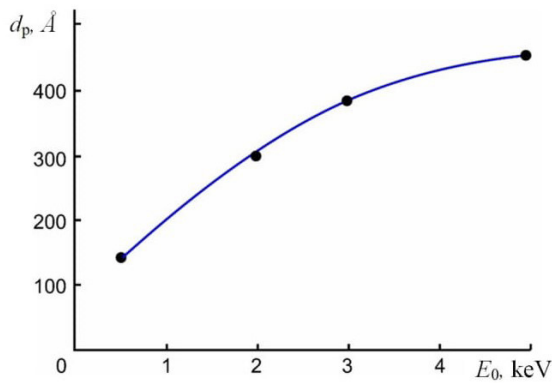


Fig. 5. A plot of the thickness of disordered layers versus the energy of Ar<sup>+</sup> ions

Table *d* increases exponentially from about 130 to 450 Å as  $E_0$  increases from 0.5 to 5 keV.

The photoelectron spectra of a CdF<sub>2</sub>/Si(111) epitaxial film before and after bombardment by Ar<sup>+</sup> ions at  $E_0 = 1$  keV with different doses, taken at  $h\nu = 21.2$  eV (Fig. 6). These spectra provide information about the density of state of the valence band electrons, and the area under the energy distribution curve is proportional to the quantum yield *Y* of photoelectrons. The top of the valence band  $E_v$  of CdF<sub>2</sub> is taken as the reference point. In the spectrum of pure CdF<sub>2</sub>, there are three clearly defined maxima (peaks) at energies  $E_{cv} = -1.6$  eV,  $-3.8$  eV, and  $-8.1$  eV. It can be assumed that peak  $E_1$  appears due to hybridization of the 5*s* level of Cd with the 2*p* level of F; the main contribution to the appearance of peak  $E_2$  is made by 5*s* levels of Cd, and that to  $E_3$  is made by 2*p* levels of F.

When CdF<sub>2</sub> film is bombarded by Ar<sup>+</sup> ions, a slight broadening of the curvilinear energy distribution of photoelectrons is observed at the dose  $D = 10^{14}$  cm<sup>-2</sup>, and a decrease in the intensity is observed at the peak of  $E_{cv} = -1.8$  and 8.1 eV. Also, the shift of the peak in the initial state of the spectrum ( $E_{cv} = 0.6 - 1.0$  eV) to the right leads to

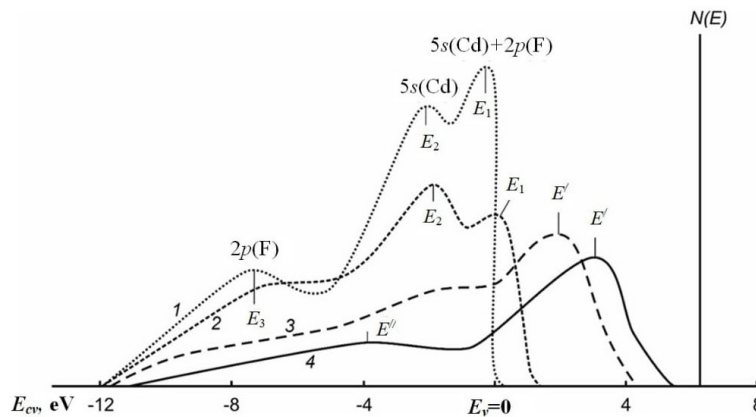


Fig. 6. The photoelectron spectra of the CdF<sub>2</sub> bombarded by Ar<sup>+</sup> ions with energy  $E_0 = 1$  keV. Doses  $D$ , cm<sup>-2</sup>: 0 (1),  $1 \cdot 10^{14}$  (2);  $5 \cdot 10^{15}$  (3),  $5 \cdot 10^{16}$  (4)

a decrease in the gap width  $E_g$  of the CdF<sub>2</sub> film. At a dose of  $5 \cdot 10^{15}$  cm<sup>-2</sup>, a new peak  $E'$  characteristic of cadmium appears instead of  $E_1$  and  $E_2$  peaks, and the value of *Y* decreases by about 2 times in this case [19, 23].

### Conclusion

In this work, the effect of bombardment with argon ions on the composition, electronic and crystal structure of the surface layers of single-crystal samples and CdF<sub>2</sub>/Si(111) films has been studied. The molecular beam epitaxial film of the CdF<sub>2</sub>/Si(111) with a thickness of 500 Å was shown to have high stoichiometric and crystalline perfection with a surface structure of  $1 \times 1$ .

It was established that the intense desorption of fluorine atoms occurred, which continued up to dose  $D = (5 - 10) \cdot 10^{16} \text{ cm}^{-2}$ . The concentration of fluorine in the surface layers were found to go down to zero in the region  $E_0 = 1 - 3 \text{ keV}$ , and not decrease to zero when  $E_0$  being greater than 4 keV. The main mechanisms of these changes were clarified.

For the first time, the thickness of layers enriched with cadmium atoms and the thickness of highly disordered  $\text{CdF}_2/\text{Si}(111)$  layers were estimated. The change in the density of state of  $\text{CdF}_2$  valence electrons upon bombardment by  $\text{Ar}^+$  ions with  $E_0 = 1 \text{ keV}$  was also studied for the first time using ultraviolet photoelectron spectroscopy (UPS) at different doses.

The results obtained in this work can undoubtedly be useful in the development of modern generation communications-electronics equipment.

## REFERENCES

1. Sugiyama M., Oshima M., MBE growth of fluorides, *Microelectron. J.* 27 (4–5) (1996) 361–382.
2. Weng J., Gao Sh.-P., Layer-dependent band gaps and dielectric constants of ultrathin fluorite crystals, *J. Phys. Chem. Solids.* 148 (Jan) (2021) 109738.
3. Illarionov Y. Y., Vexler M. I., Suturin S. M., et al., Electron tunneling in MIS capacitors with the MBE-grown fluoride layers on Si (111) and Ge (111): Role of transverse momentum conservation, *Microelectron. Eng.* 88 (7) (2011) 1291–1294.
4. Banshchikov A. G., Illarionov Yu. Yu., Vexler M. I., et al., Trends in reverse-current change in tunnel MIS diodes with calcium fluoride on Si (111) upon the formation of an extra oxide layer, *Semicond.* 53 (6) (2019) 833–837.
5. Izumi A., Kawabata K., Tsutsui K., et al., Growth of  $\text{CdF}_2\text{CaF}_2\text{Si}$  (111) heterostructure with abrupt interfaces by using thin  $\text{CaF}_2$  buffer layer, *Appl. Surf. Sci.* 104–105 (2 Sept) (1996) 417–421.
6. Sokolov N. S., Suturin S. M., MBE-growth peculiarities of fluoride ( $\text{CdF}_2\text{--CaF}_2$ ) thin film structures, *Thin Solid Films.* 367 (1–2) (2000) 112–119.
7. Mir A., Zaoui A., Bensaid D., The displacement effect of a fluorine atom in  $\text{CaF}_2$  on the band structure, *Appl. Surf. Sci.* 439 (1 May) (2018) 1180–1185.
8. Li Zh., Baskurt M., Sahin H., et al., Electronic properties of intrinsic vacancies in single-layer  $\text{CaF}_2$  and its heterostructure with monolayer  $\text{MoS}_2$ , *J. Appl. Phys.* 130 (5) (2021) 055301.
9. Kalugin A. I., Sobolev V. V., Optical properties of  $\text{CdF}_2$  in a wide energy range, *Tech. Phys.* 49 (3) (2004) 338–341.
10. Bekpulatov I. R., Imanova G. T., Kamilov T. S., et al., Formation of *n*-type CoSi monosilicide film which can be used in instrumentation, *Int. J. Modern Phys. B.* 37 (17) (2023) 22350164.
11. Bekpulatov I. R., Shomukhammedova D. S., Shukurova D. M., Ibragimova B. V., Obtaining higher manganese silicide films with high thermoelectric properties, *E3S Web Conf.* 365 (30 Jan) (2023) 05015.
12. Kamilov T. S., Rysbaev A. S., Klechkovskaya V. V., et al., The influence of structural defects in silicon on the formation of photosensitive  $\text{Mn}_4\text{Si}_7\text{--Si(Mn)--Mn}_4\text{Si}_7$  and  $\text{Mn}_4\text{Si}_7\text{--Si(Mn)--M}$  heterostructures, *Appl. Sol. Energy.* 55 (6) (2019) 380–384.
13. Rysbaev A. S., Khujaniyozov J. B., Bekpulatov I. R., et al., Effect of thermal and laser annealing on the atom distribution profiles in Si (111) implanted with  $\text{P}^+$  and  $\text{B}^+$  ions, *J. Surf. Invest.: X-ray, Synchrotron Neutron Tech.* 11 (2) (2017) 474–479.
14. Shcheulin A. S., Kupchikov A. K., Angervaks A. E., et al., Radio-frequency response of semiconducting  $\text{CdF}_2\text{:In}$  crystals with Schottky barriers, *Phys. Rev. B.* 63 (20) (2001) 205207.
15. Sorokin L. M., Kyutt R. N., Ratnikov V. V., Kalmykov A. E., Structural characterization of a short-period superlattice based on the  $\text{CdF}_2/\text{CaF}_2/\text{Si}$  (111), *Tech. Phys. Lett.* 47 (12) (2021) 893–896.
16. Ryskin A. I., Shcheulin A. S., Angervaks A. E., Semiconductor  $\text{CdF}_2\text{:Ga}$  and  $\text{CdF}_2$  in crystals as media for real-time holography, *Mater.* 5 (5) (2012) 784–817.
17. Kazansky S. A., Ryskin A. I., Clusters of group-III ions in activated fluorite-type crystals, *Phys. Solid State.* 44 (8) (2002) 1415–1425.
18. Varlamov A. G., Ulanov V. A., Donornaya provodimost' cristalla  $\text{CdF}_2\text{:V}^{3+}$ , otzhozhnogo v vakuume [Donor conductivity of the vacuum-annealed  $\text{CdF}_2\text{:V}^{3+}$  crystal], *Izvestiya Vysshikh Uchebnykh Zavedenii Rossii. Problemy Energetiki [Energy Problems]*. (9–10) (2006) 100–104 (in Russian).



19. **Rodnyĭ P. A., Khodyuk I. V., Stryganyuk G. B.**, Location of the energy levels of the rare-earth ions in  $\text{BaF}_2$  and  $\text{CdF}_2$ , *Phys. Solid State*. 50 (9) (2008) 1639–1643.
20. **Dorenbos P.**, Systematic behaviour in trivalent lanthanide charge transfer energies, *J. Phys. Cond. Matter*. 15 (49) (2003) 8417–8434.
21. **Bagraev N. T., Gimbitskaya O. N., Klyachkin L. E., et al.**, Quantum Hall effect in (cadmium fluoride)-based nanostructures, *Semicond.* 43 (1) (2009) 75–77.
22. **Umirzakov B. E., Pugacheva T. S., Tashatov A. K., Tashmukhamedova D. A.**, Electronic structure and optical properties of  $\text{CaF}_2$  films under low energy  $\text{Ba}^+$  ion-implantation combined with annealing, *Nucl. Instr. and Meth. B*. 166–167 (2 May) (2000) 572–576.
23. **Umirzakov B. E., Tashmukhamedova D. A., Muradkabilov D. M., Boltaev Kh. Kh.**, Electron spectroscopy of the nanostructures created in Si, GaAs, and  $\text{CaF}_2$  surface layers using low-energy ion implantation, *Tech. Phys.* 58 (6) (2013) 841–844.
24. **Umirzakov B. E., Tashatov A. K., Tashmukhamedova D. A., Normuradov M. T.**, Protsess formirovaniya nanoplyonok na poverkhnosti  $\text{CaF}_2$  pri ionnoy implantatsii i posleduyushchem otzhige [Nanofilm formation process on the  $\text{CaF}_2$  surface during ion implantation and subsequent annealing], *Poverkhnost'. Rentgenovskie, Sinkhronnye i Neitronnye Issledovaniya* [Surface. X-ray, Synchrotron and Neutron Investigations]. (12) (2004) 90–94 (in Russian).
25. **Umirzakov B. E., Tashmukhamedova D. A., Ruzibaeva M. K., et al.**, Investigation of change of the composition and structure of the  $\text{CdF}_2/\text{Si}$  films surface at the low-energy bombardment, *Nucl. Instr. and Meth. B*. 326 (1 May) (2014) 322–325.
26. **Tashmukhamedova D. A., Yusupjanova M. B.**, Emission and optical properties of  $\text{SiO}_2/\text{Si}$  films, *J. Surf. Invest.: X-ray, Synchrotron Neutron Tech.* 10 (6) (2016) 1273–1275.
27. **Abduvayitov A. A., Boltaev Kh. Kh., Rozikov G. A.**, Study of the composition of uncontrolled impurities and the profiles of their distribution at the Ni–CdS interface, *J. Surf. Invest.: X-ray, Synchrotron Neutron Tech.* 16 (5) (2022) 860–863.

#### СПИСОК ЛИТЕРАТУРЫ

1. **Sugiyama M., Oshima M.** MBE growth of fluorides // *Microelectronics Journal*. 1996. Vol. 27. No. 4–5. Pp. 361–382.
2. **Weng J., Gao Sh.-P.** Layer-dependent band gaps and dielectric constants of ultrathin fluorite crystals // *Journal of Physics and Chemistry of Solids*. 2021. Vol. 148. January. P. 109738.
3. **Illarionov Y. Y., Vexler M. I., Suturin S. M., Fedorov V. V., Sokolov N. S., Tsutsui K., Takahashi K.** Electron tunneling in MIS capacitors with the MBE-grown fluoride layers on Si (111) and Ge (111): Role of transverse momentum conservation // *Microelectronic Engineering*. 2011. Vol. 88. No. 7. Pp. 1291–1294.
4. **Банщиков А. Г., Илларионов Ю. Ю., Векслер М. И., Wachter S., Соколов Н. С.** Характер изменения обратного тока в туннельных МДП-диодах с фторидом кальция на Si (111) при создании дополнительного оксидного слоя // *Физика и техника полупроводников*. 2019. Т. 53. № 6. С. 844–849.
5. **Izumi A., Kawabata K., Tsutsui K., Sokolov N. S., Novikov S. V., Khilko A. Yu.** Growth of  $\text{CdF}_2/\text{CaF}_2/\text{Si}(111)$  heterostructure with abrupt interfaces by using thin  $\text{CaF}_2$  buffer layer // *Applied Surface Science*. 1996. Vol. 104–105. 2 September. Pp. 417–421.
6. **Sokolov N. S., Suturin S. M.** MBE-growth peculiarities of fluoride ( $\text{CdF}_2$ – $\text{CaF}_2$ ) thin film structures // *Thin Solid Films*. 2000. Vol. 367. No. 1–2. Pp. 112–119.
7. **Mir A., Zaoui A., Bensaïd D.** The displacement effect of a fluorine atom in  $\text{CaF}_2$  on the band structure // *Applied Surface Science*. 2018. Vol. 439. 1 May. Pp. 1180–1185.
8. **Li Zh., Baskurt M., Sahin H., Gao S., Kang J.** Electronic properties of intrinsic vacancies in single-layer  $\text{CaF}_2$  and its heterostructure with monolayer  $\text{MoS}_2$  // *Journal of Applied Physics*. 2021. Vol. 130. No. 5. P. 055301.
9. **Калугин А. И., Соболев В. В.** Оптические свойства  $\text{CdF}_2$  в широкой области энергии // *Журнал технической физики*. 2004. Т. 74. № 3. С. 58–61.
10. **Bekpulatov I. R., Imanova G. T., Kamilov T. S., Igamov B. D., Turapov I. Kh.** Formation of *n*-type CoSi monosilicide film which can be used in instrumentation // *International Journal of Modern Physics B*. 2023. Vol. 37. No. 17. P. 22350164.

11. **Bekpulatov I. R., Shomukhamedova D. S., Shukurova D. M., Ibragimova B. V.** Obtaining higher manganese silicide films with high thermoelectric properties // *E3S Web of Conferences*. 2023. Vol. 365. 30 January. Article No. 05015 (7 p).
12. **Kamilov T. S., Rysbaev A. S., Klechkovskaya V. V., Orekhov A. S., Igamov B. D., Bekpulatov I. R.** The influence of structural defects in silicon on the formation of photosensitive  $Mn_4Si_7$ -Si(Mn)- $Mn_4Si_7$  and  $Mn_4Si_7$ -Si(Mn)-M heterostructures // *Applied Solar Energy*. 2019. Vol. 55. No. 6. Pp. 380–384.
13. **Рысбаев А. С., Хужаниязов Ж. Б., Бекпулатов И. Р., Рахимов А. М., Пардаев О. Р.** Исследование влияния термического и лазерного отжига на профили распределения атомов в Si (111), имплантированного ионами  $P^+$  и  $B^+$  // *Поверхность. Рентгеновские, синхротронные и нейтронные исследования*. 2017. № 4. С. 98–103.
14. **Shcheulin A. S., Kupchikov A. K., Angervaks A. E., Onopko D. E., Ryskin A. I., Ritus A. I., Pronin A. V., Volkov A. A., Lunkenheimer P., Loidl A.** Radio-frequency response of semiconducting  $CdF_2$ : In crystals with Schottky barriers // *Physical Review B*. 2001. Vol. 63. No. 20. P. 205207.
15. **Сорокин Л. М., Кютт Р. Н., Ратников В. В., Калмыков А. Е.** Структурная характеристика короткопериодной сверхрешетки на основе гетероструктуры  $CdF_2/CaF_2/Si(111)$  методами просвечивающей электронной микроскопии и рентгеновской дифрактометрии // *Письма в Журнал технической физики*. 2021. Т. 47. № 15. С. 3–6.
16. **Ryskin A. I., Shcheulin A. S., Angervaks A. E.** Semiconductor  $CdF_2$ : Ga and  $CdF_2$  in crystals as media for real-time holography // *Materials (Basel)*. 2012. Vol. 5. No. 5. Pp. 784–817.
17. **Казанский С. А., Рыскин А. И.** Кластеры ионов III группы в активированных кристаллах типа флюорита // *Физика твердого тела*. 2002. Т. 44. № 8. С. 1356–1366.
18. **Варламов А. Г., Уланов В. А.** Донорная проводимость кристалла  $CdF_2:V^{3+}$ , отожженного в вакууме // *Известия высших учебных заведений. Проблемы энергетики*. 2006. № 9–10. С. 100–104.
19. **Родный П. А., Ходюк И. В., Стрыганюк Г. Б.** Энергетическое положение редкоземельных ионов в  $BaF_2$  и  $CdF_2$  // *Физика твердого тела*. 2008. Т. 50. № 9. С. 1578–1581.
20. **Dorenbos P.** Systematic behaviour in trivalent lanthanide charge transfer energies // *Journal of Physics: Condensed Matter*. 2003. Vol. 15. No. 49. Pp. 8417–8434.
21. **Баграев Н. Т., Гимбицкая О. Н., Клячкин Л. Е., Маляренко А. М., Шелых И. А., Рыскин А. И., Щеулин А. С.** Квантовый эффект Холла в наноструктурах на основе фторида кадмия // *Физика и техника полупроводников*. 2009. Т. 43. № 1. С. 844–849.
22. **Umirzakov V. E., Pugacheva T. S., Tashatov A. K., Tashmukhamedova D. A.** Electronic structure and optical properties of  $CaF_2$  films under low energy  $Ba^+$  ion-implantation combined with annealing // *Nuclear Instruments and Methods in Physics Research B*. 2000. Vol. 166–167. 2 May. Pp. 572–576.
23. **Умирзаков Б. Е., Ташмухамедова Д. А., Мурадкбиллов Д. М., Болтаев Х. Х.** Электронная спектроскопия наноструктур, созданных в поверхностных слоях Si, GaAs и  $CaF_2$  методом низкоэнергетической ионной имплантации // *Журнал технической физики*. 2013. Т. 83. № 6. С. 66–70.
24. **Умирзаков Б. Е., Ташатов А. К., Ташмухамедова Д. А., Нормурадов М. Т.** Процесс формирования нанопленок на поверхности  $CaF_2$  при ионной имплантации и последующем отжиге // *Поверхность. Рентгеновские, синхротронные и нейтронные исследования*. 2004. № 12. С. 90–94.
25. **Umirzakov V. E., Tashmukhamedova D. A., Ruzibaeva M. K., Djurabekova F. G., Danaev S. B.** Investigation of change of the composition and structure of the  $CdF_2/Si$  films surface at the low-energy bombardment // *Nuclear Instruments and Methods in Physics Research B*. 2014. Vol. 326. 1 May. Pp. 322–325.
26. **Ташмухамедова Д. А., Юсупжанова М. Б.** Эмиссионные и оптические свойства тонких пленок  $SiO_2/Si$  // *Поверхность. Рентгеновские, синхротронные и нейтронные исследования*. 2016. № 12. С. 89–91.
27. **Abduvayitov A. A., Boltaev Kh. Kh., Rozikov G. A.** Study of the composition of uncontrolled impurities and the profiles of their distribution at the Ni–CdS interface // *Journal of Surface Investigation: X-ray, Synchrotron and Neutron Techniques*. 2022. Vol. 16. No. 5. Pp. 860–863.



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